IMPROVING THE FUNCTION FOR GRAIN BOUNDARY ENERGY INTERPOLATION IN URANIUM DIOXIDE

by

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A senior thesis submitted to the faculty of

Brigham Young University - Idaho

in partial fulfillment of the requirements for the degree of

Bachelor of Science

Department of Physics

Brigham Young University - Idaho

December 2016
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DEPARTMENT APPROVAL

of a senior thesis submitted by

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Others have made efforts to find an interpolation function for the grain boundary (GB) energies of uranium dioxide, based on work done by Bulatov et al. [Acta Mater. 65, 161 (2014)]. This work developed a MATLAB® script based on Harbison [B.S. Thesis, Brigham Young University - Idaho (2015)] and Bulatov et al. to improve such a function. This work collected molecular dynamics data using the LAMMPS (Large-scale Atomic/Molecular Massively Parallel Simulation) program developed at Sandia National Laboratory. This work collected results for the ⟨100⟩, ⟨110⟩, and ⟨111⟩ symmetric tilt and twist GBs. Calculating the new data with an 800 K anneal allowed the atoms to relax to a lower energy state. The ⟨100⟩ and ⟨110⟩ symmetric tilt and ⟨110⟩ twist sets show an improved fit, whereas the ⟨100⟩ twist and ⟨111⟩ symmetric tilt and twist sets show unexpected trends. Further research needs to be done for the ⟨100⟩ and ⟨111⟩ sets to determine why the fitting procedure does not accurately reflect the expected results. Additional research should also be done to determine if outlying data points necessitate fitting additional cusps.
I would first like to thank the Department of Energy Office of Science for the Visiting Faculty Program (VFP) - Student, and the Nuclear Energy Advanced Modeling and Simulation (NEAMS) programs which allowed me this research opportunity. Thanks also goes to Brigham Young University - Idaho and Idaho National Laboratory for providing me with the necessary facilities to do this work. I would especially like to thank my mentor, Dr. Yongfeng Zhang, for his patience and guidance as I have worked on this research. Additionally I would like to thank Dr. Evan Hansen, John-Michael Bradley, and Dr. Xianming Bai for their valuable contributions to my understanding. Thanks also goes to my committee who helped me to clarify my words and ideas to make them clear for this thesis. Finally, I would like to thank my wife Katrina who has constantly supported and strengthened me as I have spent so much time doing this work.
Contents

Table of Contents xi
List of Figures xiii
1 Introduction 1
2 Background 3
3 Methods 7
  3.1 Molecular Dynamics ........................................ 7
  3.2 Bulatov et al.’s Methods .................................... 9
  3.3 Code Analysis .............................................. 10
    3.3.1 The Fitting Code .................................... 11
    3.3.2 The Energy Calculation Code .......................... 12
  3.4 Reduced Chi-Square Statistic ............................... 13
    3.4.1 Developing the P and Q Matrices ....................... 13
    3.4.2 Calculating Reduced Chi Squared ...................... 18
4 Results and Discussion 21
  4.1 Validation of P and Q Matrices ............................ 21
  4.2 Fitting Results .......................................... 21
  4.3 Reduced Chi-square Results .............................. 24
5 Conclusion 29
Bibliography 31
A List of Parameters 35
B Grain Boundary Representations 39
  B.1 Axis-Angle Representation .................................. 39
  B.2 Rodrigues Representation .................................. 40
  B.3 Fundamental Zone Representation ........................ 41
C Graphs 43
D Orientation Matrix Generator 47
E  Rotation Matrix Generator  61

F  genOrientationMatrix.sh Bash Script  69
# List of Figures

1.1 Example of the fluorite crystal structure. .................................. 1

2.1 Examples and types of grain boundaries. ................................. 4

3.1 An example of crystal structure after annealing. ....................... 8

3.2 The theoretical relationship between high-symmetry subsets and fundamental zone. .............................................................. 10

3.3 The general form of an RSW function. .................................... 11

3.4 Geometric method of determining grain boundary normals. .......... 16

4.1 A comparison of the $\langle 100 \rangle$ copper curve with the calculated results. ...... 22

4.2 Results for the $\langle 100 \rangle$ fitting. ........................................ 24

4.3 Results for the $\langle 110 \rangle$ fitting. ........................................ 24

4.4 Results for the $\langle 111 \rangle$ fitting. ........................................ 25

4.5 Comparison of the PQ matrices with the expected result for $\langle 100 \rangle$ tilt. .. 25

4.6 Possible changes to fitting functions for the 1D twist subsets. .......... 26

C.1 A comparison of the $\langle 100 \rangle$ copper curves with the calculated results. .. 43

C.2 A comparison of the $\langle 110 \rangle$ copper curves with the calculated results. .. 44

C.3 A comparison of the $\langle 111 \rangle$ copper curves with the calculated results. .. 44

C.4 Comparison of the PQ matrices with the expected result for $\langle 100 \rangle$ 1D subset. 45

C.5 Comparison of the PQ matrices with the expected result for $\langle 110 \rangle$ 1D subset. 45

C.6 Comparison of the PQ matrices with the expected result for $\langle 111 \rangle$ 1D subset. 46
Chapter 1

Introduction

Today’s nuclear reactors primarily use uranium dioxide (UO$_2$) as their fuel source. Understanding the properties of UO$_2$ requires an analysis of the basic crystal structure of the material. The ceramic material UO$_2$ has a series of crystal lattices joined together in various ways (called twist, tilt, or mixed boundaries) to create it. This material has a fluorite crystal structure, where the uranium atoms form a face-centered cubic (fcc) lattice, and the oxygen atoms form a simple cubic lattice within the fcc frame (see Figure 1.1).

Understanding the various properties of the fuel allows nuclear reactors to run as safely and effectively as possible. A few of these properties include thermal conductivity (how well heat flows through the material), fission gas release (how some of the fission products move throughout the material as gases), and mechanical stability (i.e. how the material bends or cracks under pressure or heat). Taking thermal conductivity as an example, knowledge of this material property allows the most effective

![Figure 1.1](image_url) An image representing the fluorite crystal structure. For UO$_2$, the smaller spheres indicate the uranium atoms, and the larger spheres indicate the oxygen atoms. Image courtesy of the University of Cambridge under the Creative Commons license.
use of coolant to keep the reactor within operating temperatures, maximizing both efficiency and safety. Knowledge of other material properties allows for similar gains in efficiency, safety, or both.

Interest in understanding UO$_2$ while in-reactor has led to efforts to more deeply understand the properties of the material. Currently, Idaho National Laboratory (INL) faces the challenge of not having a completely accurate model of grain boundary energy anisotropy. The current model assumes an isotropic energy, and this leads to an inability to model the material parameters correctly while in-reactor. This in turn makes efforts in nuclear energy less efficient and/or safe than it otherwise could be. INL aims to accurately model nuclear fuel while in-reactor, allowing accurate predictions regarding how the material properties will change.

This work adds to the safety and efficiency of using nuclear energy by providing the necessary information to accurately calculate the material properties of UO$_2$ in-reactor. Specifically, this work improves the fitting parameters for grain boundary (GB) energy interpolation for UO$_2$ by using molecular dynamics (MD) results calculated by Zhang$^2$ and Hansen$^3$ with an anneal of 800 K. Furthermore, this work begins preliminary efforts towards using more accurate functions to describe GB energy behavior. Previous data did not use annealed crystal structures,$^4$ which prevented the atoms from finding their ideal energetic minimum. The 800 K anneal allows the atoms to relax to a value closer to their global minimum, as shown in Chapter 4. A database will store these simulated energies, and a MATLAB® script will use the database to fit the function parameters. INL will incorporate the updated parameters into its mesoscale phase field modeling platform MARMOT for use in modeling nuclear fuels. As the modeling software incorporates these parameters, various tests of the UO$_2$ fuel can determine how the material properties change while in-reactor.
Chapter 2

Background

Tiny crystals called grains make up every polycrystalline materials (such as ceramics). The orientation of each grain does not generally depend on the orientation of the surrounding grains. Therefore, depending on how the crystal formed, the crystal structures will possibly not line up at the interfaces where two grains meet. This “atomic mismatch” leads to broken or stretched atomic bonds where atoms will not line up relative to a perfect crystal structure, creating defects called grain boundaries (GBs, see Figure 2.1a). The most popular way to parameterize a GB uses the five degree-of-freedom (DoF) model. This model only uses the macroscopic DoFs (the observable DoFs corresponding to the misorientation and inclination), ignoring the three translational DoFs (the ability of the grain to move or slide anywhere in space) possessed by each grain. Three of the five DoFs specify the misorientation (or misalignment) of the grains with respect to each other. The other two DoFs specify the orientation of the grain boundary plane (called the inclination). The rotation axis and angle define the misorientation DoFs, and the GB normal defines the inclination DoFs.

Three specific types of GBs occur in polycrystalline materials: twist, tilt, and mixed GBs. These GBs describe the misorientation of two grains with respect to each other. Twist boundaries have the axis of rotation between the two grains and the GB normal parallel to each other. Tilt boundaries can be either symmetric or asymmetric. A perpendicular
A tilt boundary results from the relative rotation of two crystal grains. The tilt boundary is defined by a boundary plane, where the atoms on one side of the plane mirror the other side, creating a symmetric tilt boundary. Asymmetric tilt boundaries have unequal angles. Figure 2.1b shows a representation of tilt boundaries (top) and twist boundaries (bottom). A mixed GB combines twist and tilt boundary characteristics.

**Figure 2.1** A representation of GBs, where (a) shows an example of a grain boundary and (b) shows an example of GB types. In (a) circles represent individual atoms of the grains, and the line represents the grain boundary. Atomic mismatch between the differently oriented grains causes an excess of energy within the material, which has an effect on the material’s properties. Image courtesy of the University of Cambridge under the Creative Commons license. In (b) the tilt GB (top) has a perpendicular relationship between the axis of rotation and the GB normal, while the twist GB (bottom) has a parallel relationship between the two. Image courtesy of Wikipedia under the Creative Commons license.

GBs have various effects on material properties, making them important to understand.\textsuperscript{6,8,9} The crystal structure has extra energy because of the atomic mismatch at the boundaries. This extra energy, called GB energy, gives rise to GB motion. Knowing and predicting how the GBs will move allows for more accurate calculations of a material’s properties. Thus, GB energy needs to be understood to accurately model the evolution of material
properties.

Two methods, isotropic and anisotropic, model GB energy (among others). The isotropic model most commonly (and most easily) describes GB energy. This model ignores the impact of inclination on the GB energy, and assumes equal inclinations for a given misorientation, reducing the five-dimensional (5D) parameter space to a three-dimensional (3D) parameter space. Historically, isotropic models assumed that the inclination had little or no impact on the GB energy. Later, researchers used this model because of the difficulty in creating a full five DoF model.\(^8\) Alternatively, the anisotropic approach seeks to quantify the effect that misorientation \textit{and} inclination have on the GB energy. Currently, researchers acknowledge the need for a full five DoF model of GB energy, but assert the difficulty inherent in developing such a model.\(^7,8,10\) Despite these difficulties, GB energy functions for certain materials, namely fcc metals copper, gold, aluminum, and nickel, have proven successful.\(^9\) Recently,\(^4\) similar efforts created a GB energy function for UO\(_2\). This work improves the accuracy of that energy interpolation function.
Chapter 3

Methods

Any fitting procedure requires a sufficient amount of data, but unfortunately such data does not exist for uranium dioxide (UO$_2$) in the literature. As a work-around, this work used molecular dynamics (MD) simulations$^2,3$ as fitting data to calculate the GB energies of various lattices based on the coincident site lattice (CSL) model. This model builds off of the idea that the GB energy has lower values when more lattice sites coincide. A number defined as the Σ-number describes the number of coincident sites per total number of lattice sites in a given unit cell of a crystal.$^7,10$ This work developed a MATLAB® script using Bulatov et al.’s methods$^9$ and building off of Harbison’s script$^4$ to fit parameters to the gathered data. A reduced chi-square statistic determined the effectiveness of the fit.

3.1 Molecular Dynamics

Zhang$^2$ and Hansen$^3$ collected simulation results from the Large-scale Atomic/Molecular Massively Parallel Simulation (LAMMPS) software (developed at Sandia National Laboratory$^{11}$) for a number of twist, tilt, and mixed GBs. They performed these calculations by simulating two crystals of UO$_2$ and placing them together in various orientations. A GB forms at the interface, creating GB energy. Calculating the energy of the system from the interatomic forces inside the crystal, and comparing that energy to the energy of a single grain
(of the same size as the combined two grains) determines the energy at the GB. Calculating the GB energy follows the form:

\[ E_{\text{GB}} = \frac{|E_{\text{single grain}} - E_{\text{two grains}}|}{2A_{\text{GB}}} \]  

Here, \( E_{\text{GB}} \) represents the energy at the grain boundary, and \( E_{\text{single grain}} \) and \( E_{\text{two grains}} \) represent the energies of the single and double grains respectively. \( A_{\text{GB}} \) represents the area of the grain boundary. Figure 3.1 shows an example of how the atoms align. Harbison’s original calculations used no anneal (\( T_{\text{max}} \approx 0 \) K), only allowing the atoms to relax to their local minima. This work used an anneal of 800 K, allowing the atoms to relax to a better estimate of their global minimum value as shown in Chapter 4. This work used the same misorientation angles for the GB energy calculations that Harbison used. The fitting procedure uses these energies to produce parameters describing the five-dimensional GB space.

\[ E_{\text{GB}} = \frac{|E_{\text{single grain}} - E_{\text{two grains}}|}{2A_{\text{GB}}} \]  

Figure 3.1 These figures demonstrate example crystal structures of UO\textsubscript{2} after an annealing process. The better the atoms line up, the lower the energy. (a) shows an example of a mostly aligned GB, indicative of a lower energy. (b) shows an example of a misaligned GB, indicative of a higher energy. These two images show results from a \( \langle 111 \rangle \) twist simulation. Different viewpoints show different amounts of alignment. The LAMMPS simulation package takes care of all the calculations to determine the energy at these GBs. Images courtesy of Dr. Evan Hansen, used with permission.
3.2 Bulatov et al.’s Methods

This work implemented Bulatov et al.’s hierarchical interpolation method to find the energy of an arbitrary GB in the five-space. They chose three three-dimensional (3D) axes with at least two-fold symmetry (called high-symmetry axes) to use as scaffolding to build the entire five-dimensional (5D) function. Bulatov et al. and this work chose the \( \langle 100 \rangle \), \( \langle 110 \rangle \), and the \( \langle 111 \rangle \) sets for their four-, two-, and three-fold rotational symmetries respectively.* Each 3D subset builds from interpolation of its own one- and two-dimensional subsets. The symmetric tilt and twist GBs for each set were fitted first because of their simplicity. The rotation angle fully defines the energies for these subsets, making them one-dimensional (in Figure 3.2a, the darker bands in the smaller circles). Having the parameters from the symmetric tilt subset allows interpolation to the asymmetric, or general, tilt subset. A second rotation angle defining the rotation of the second grain makes this subset two-dimensional (the lighter, wider band around the symmetric subset). A combination of the general tilt (two dimensions) and the twist subsets (one dimension) interpolates the 3D subset for each high-symmetry axis (the three smaller circles). Bulatov et al. and this work used these three 3D subsets to interpolate the GB 5D space. Figure 3.2b shows the simplified GB space using the Rodrigues fundamental zone representation. Appendix B provides a further explanation of Rodrigues space and the fundamental zone.

Bulatov et al. and this work used the Read-Shockley-Wolf (RSW) functions,\(^{14}\) which take the form:

\[
E_{\text{min}} + (E_{\text{max}} - E_{\text{min}}) \sin \left( \frac{\pi}{2} \frac{\theta - \theta_{\text{min}}}{\theta_{\text{max}} - \theta_{\text{min}}} \right) \left( 1 - a \log \left( \sin \left( \frac{\pi}{2} \frac{\theta - \theta_{\text{min}}}{\theta_{\text{max}} - \theta_{\text{min}}} \right) \right) \right),
\]

where \( \theta \) is the misorientation angle, \( \theta_{\text{min}} \) and \( \theta_{\text{max}} \) represent the minimum and maximum angles on the domain respectively, \( a \) is a shaping parameter, and \( E_{\text{min}} \) and \( E_{\text{max}} \) represent

*Symmetry operations for cubic crystals include rotating by 90°, 180°, or 120° about any \( \langle 100 \rangle \), \( \langle 110 \rangle \), or \( \langle 111 \rangle \) axis respectively.\(^{13}\) Thus, the \( \langle 100 \rangle \) set has four-fold symmetry (360°/90° = 4), the \( \langle 110 \rangle \) set has two-fold symmetry (360°/180° = 2), and the \( \langle 111 \rangle \) set has three-fold symmetry (360°/120° = 3).
the energy at $\theta_{\text{min}}$ and $\theta_{\text{max}}$ respectively. Each RSW function covers a “low-angle” subset (around 15°, with larger domains being less accurate)\(^\text{10,14}\) of the domain in the 1D GB space. Figure 3.3 shows an example of a simple RSW function. Stitching together multiple RSW functions forms the 1D subsets.

**Figure 3.2** Figure 2 from Bulatov et al.\(^\text{9}\) (a) demonstrates the theoretical relationship between the high-symmetry subsets of the 5D GB space. Each multi-dimensional subset interpolates from smaller-dimensional subsets. (b) shows the Rodrigues space representation of the fundamental zone of all GBs as built from three high-symmetry axes (\((100)\), \((110)\), and \((111)\)). The unit vectors along the axis identify the boundary plane inclination in the frame of grain one. A parallel vector thus represents a twist boundary, a perpendicular vector represents a tilt boundary, and neither parallel nor perpendicular vectors represent a mixed boundary.

### 3.3 Code Analysis

Harbison\(^\text{4}\) and Bulatov et al.\(^\text{9}\) developed MATLAB\textsuperscript{®} scripts for their work. This work analyzed these codes and used the ideas from them to develop the code that generated the parameters listed in Appendix A.
3.3 Code Analysis

Figure 3.3 An example of an RSW function with $\theta_{\text{min}} = 0^\circ$, $\theta_{\text{max}} = 15^\circ$, and $a$ (the shaping parameter) = 0.5. Combining these functions into a Piecewise set over a given domain gives the GB energy curves their distinct, cusp-like behavior. The RSW functions scale based on $E_{\text{min}}$ and $E_{\text{max}}$. In this example, $E_{\text{min}} = 0$ and $E_{\text{max}} = 1$. Note that $E_{\text{min}}$ and $E_{\text{max}}$ do not represent the lowest and highest energies in the domain, but rather represent the energy at $\theta_{\text{min}}$ and $\theta_{\text{max}}$ respectively, meaning that the value of $E_{\text{min}}$ could be higher than the value of $E_{\text{max}}$.

3.3.1 The Fitting Code

This work performed an extensive analysis of Harbison’s code to learn how it works and to implement the ideas therein. The basic outline for the fitting procedure follows. First, a database containing energies associated with either a twist or tilt GB on one of the three high-symmetry axes provides the fitting data. A separate database provides the test parameters which define starting points for the fitted parameters. The parameters found from the 1D fits assist in fitting the higher-dimensional sets. Important angles specify where to expect low energies, such as the $\Sigma5$ boundary for the $\langle 100 \rangle$ symmetric tilt subset. The $\Sigma$-number from CSL theory determines the angles. Because the $\Sigma$-number designates the number of
lattice points between each coincident site (and assuming the separation distance between each lattice site, or the lattice constant is known) the angle of the GB misorientation can be determined. A value known as the $e_{RGB}$ parameter scales every energy in the parameter vector to minimize the potential for error in calculations. The $e_{RGB}$ parameter represents the energy of an arbitrary, random GB, and represents an average of the material’s GB energies. Thus, making relevant comparisons requires unscaling the energies based on the units of energy desired (typically J/m²). All of the parameters and the angle-energy pairs from the database get passed into a grid-search fitting function. This work gave each subset a different initial step size to avoid a numerical error where the steps would take the angles currently being looked at outside of their domain. Without this, the grid-search procedure would not return the correct amount of values, preventing the code from running to completion.

After fitting the six one-dimensional subsets and the three two-dimensional subsets, interpolating the twist (1D) and asymmetric tilt (2D) subsets calculates the mixing parameters to fit the three-dimensional subsets. The mixing parameters define the relationship between the twist and general tilt subsets within a high-symmetry axis - i.e. the relationship between the small dark bands representing the twist boundaries and the lighter, wider bands representing the tilt boundaries in Figure 3.2a. The final step calculates the weighting parameters, which defines the relationship between the three high-symmetry subsets. Equations defining the various relationships can be found in Bulatov et al.’s work.9

### 3.3.2 The Energy Calculation Code

Bulatov et al.’s open-source MATLAB® code,9 GB5DOF.m, calculates the energy of an arbitrary GB in certain fcc metals. This work uses this script for calculating an arbitrary GB in UO$_2$, and proceeds as follows. First, it compares all symmetrically equivalent representations of a GB (on a per-axis basis) to calculate metrics defining the “distance” between the GB and all three high-symmetry axes (for cubic crystals, there are 24 equivalent representations$^{13}$). Because of the three, six, and four unique axes for the $\langle 100 \rangle$, $\langle 110 \rangle$, and $\langle 111 \rangle$ sets respectively,
the script calculates a maximum of $6 \times 24 = 144$ distances, afterwards discarding any that exceed a predefined cutoff distance. After calculating all distances, the script keeps only the unique representations to avoid double-counting.\footnote{3} It then calculates energies for each unique distance in each subset, then weights and sums them to give the interpolated energy for the specified GB.

### 3.4 Reduced Chi-Square Statistic

A good way to test how well a function fits the data uses a reduced chi-square goodness-of-fit statistic.\footnote{15} Bulatov et al.’s function required the orientation matrices (which Bulatov et al. calls the P and Q matrices for the first and second grains respectively) as input parameters to calculate this statistic. These three by three matrices specify the orientation in a lab frame of the two grains individually. A good fit will have a reduced chi-square value close to one, while those values greater than one indicate an under fit, and those values less than one indicate an over fit.\footnote{15}

#### 3.4.1 Developing the P and Q Matrices

This work created the P and Q matrices. Because of the vast quantity of work done with crystallography over the past few decades, many different methods can specify the orientation matrices of grains. A rotation matrix also needed to be calculated which rotates the axis of rotation to the $[100]$ direction, as Bulatov et al.’s energy calculation code assumes. This work used three methods, following the method prescribed in MARMOT, using the Rodrigues rotation formula, and using the Bunge rotation matrix, in the process of developing these matrices.
MARMOT Method

MARMOT, Idaho National Laboratory’s (INL’s) mesoscale phase-field modeling platform,\textsuperscript{16} calculates the P and Q matrices for the grains using Euler angles as input parameters. MARMOT uses the Bunge convention to convert the Euler angles to orientation matrices. The Bunge convention uses the $ZXZ$ or $ZX'Z''$ rotation set, which rotates first about the $z$ axis, then the rotated $x$ axis, and finally about the rotated $z$ axis. Multiplying the $z$, $x$ and $z$ rotation matrices together in that order generates the formula to convert from Bunge Euler angles to the rotation matrix:

\[
\begin{pmatrix}
   c_1 & c_3 - c_2 & s_1 & s_3 & -c_1 & s_3 - c_2 & c_3 & s_1 & s_1 & s_2 \\
   c_3 & s_1 + c_1 & c_2 & s_3 & c_1 & c_2 & c_3 & -s_1 & s_3 & -c_1 & s_2 \\
   s_2 & s_3 & c_3 & s_2 & c_2 &
\end{pmatrix}
\] (3.3)

where $c_n$ and $s_n$ represent the cosine and sine of the respective angles (1 represents the first $z$ rotation, 2 represents the $x$ rotation, and 3 represents the second $z$ rotation, usually referred to as\textsuperscript{17} $\varphi_1$, $\Phi$, $\varphi_2$).

MARMOT calculates the rotation matrix using the GB normal by finding the rotation matrix required to rotate that vector to the [100] direction. In MARMOT, input files set up the simulations. In the input files different sections (called blocks) specify material parameters, boundary conditions, initial conditions, and the physical models to use to solve the problem (among others). A horizontal or vertical boundary for tilt or twist GBs respectively defined the initial condition used to calculate the rotation matrices in MARMOT for this set of problems. Because of this set up, tilt boundaries had GB normals along the [010] axis, and twist boundaries had GB normals along the [100] axis.
Rodrigues Rotation Formula

The Rodrigues rotation formula\(^\PageIndex{18}\) (RRF) calculates the rotation matrices given an axis and an angle using the following formula:

\[ \mathbf{R} = \mathbf{I} + \sin \theta \, \mathbf{K} + (1 - \cos \theta) \, \mathbf{K}^2, \quad (3.4) \]

where \( \mathbf{I} \) is the 3x3 identity matrix, \( \theta \) is the angle rotated through, and \( \mathbf{K} \) is the skew-symmetric matrix formed by the axis of rotation (\( \mathbf{a} \), where \( \mathbf{a} \) has components \( a_x \), \( a_y \), and \( a_z \)) by:

\[
\begin{bmatrix}
0 & -a_z & a_y \\
-a_z & 0 & -a_x \\
a_y & a_x & 0 \\
\end{bmatrix}, \quad (3.5)
\]

This work calculated the rotation matrices two different ways with this orientation matrix formulation. The first method used the MARMOT-generated rotation matrices. A second method calculated the rotation matrices using geometric arguments (see Figure 3.4). From the geometric arguments this work identified the normals given in Table 3.1. Appendix E shows the code used to generate the rotation matrices.

Bunge Rotation Matrix

MARMOT uses the Bunge rotation matrix (see Equation (3.3)) to create the orientation matrices. This work used various methods to calculate the Euler angles, of which three are briefly described here. The first two methods use the Euler angles to calculate the entirety of the rotation matrix.

First, this work tried to use scripts developed to calculate the various Euler angles for MARMOT. These scripts did not work because of the same assumptions made earlier about the orientation of the GB, namely, that all pure tilt GBs have a normal of \([010]\), and that all pure twist GBs have a normal of \([\bar{1}00]\). This work assumed boundary conditions to be either perpendicular or parallel to the rotation axis, while MARMOT’s boundary conditions assume GB normals along the \(x\)- or \(y\)-axes.
Chapter 3 Methods

Figure 3.4 A geometric method of determining the normals of a GB. (a) to (c) show the normals for a GB perpendicular to the axis of rotation (a twist GB). The axis about which the grains rotate defines the GB normal. (d) and (e) show the normals for a GB parallel to the axis of rotation (a tilt GB). The same GB normal for \( \langle 110 \rangle \) tilt boundaries can be used for \( \langle 111 \rangle \) tilt boundaries.

The second method used an open-source MATLAB\textsuperscript{®} package called MTEX.\textsuperscript{19} This package calculates Euler angles using quaternions. These Euler angles did not generate the correct results either, for the most part creating the same sorts of graphs as the MARMOT method.

The working method used the mathematics of quaternions directly.\textsuperscript{20} This work calculated the quaternions based on the misorientation axis and angle. A quaternion is a four-dimensional vector containing one real part, and three imaginary parts, calculated as follows:

$$q = \left[ \cos \left( \frac{\theta}{2} \right), a_x \sin \left( \frac{\theta}{2} \right), a_y \sin \left( \frac{\theta}{2} \right), a_z \sin \left( \frac{\theta}{2} \right) \right],$$  \hspace{1cm} (3.6)

with axis \( a \) and misorientation angle \( \theta \). After converting the axis and misorientation angle to a quaternion, another conversion changes the quaternion to a set of Bunge Euler angles. Calculation of the angles uses Python’s \texttt{atan2()} method, allowing all four quadrants in
Table 3.1 Table of GB normals for different GB types. The normalized dot product of the axis with the GB normal is zero in all tilt cases and one in all twist cases. Each subset has two options for the grain boundary normals because of inversion symmetries.

<table>
<thead>
<tr>
<th>Axis</th>
<th>Boundary Type</th>
<th>GB Normal</th>
</tr>
</thead>
<tbody>
<tr>
<td>\langle 100 \rangle</td>
<td>Tilt</td>
<td>[010]</td>
</tr>
<tr>
<td></td>
<td></td>
<td>[0\bar{1}0]</td>
</tr>
<tr>
<td>\langle 110 \rangle</td>
<td>Tilt</td>
<td>[1\bar{1}0]</td>
</tr>
<tr>
<td></td>
<td></td>
<td>[\bar{1}10]</td>
</tr>
<tr>
<td>\langle 111 \rangle</td>
<td>Tilt</td>
<td>[110]</td>
</tr>
<tr>
<td></td>
<td></td>
<td>[\bar{1}10]</td>
</tr>
<tr>
<td>\langle 100 \rangle</td>
<td>Twist</td>
<td>[100]</td>
</tr>
<tr>
<td></td>
<td></td>
<td>[\bar{1}00]</td>
</tr>
<tr>
<td>\langle 110 \rangle</td>
<td>Twist</td>
<td>[110]</td>
</tr>
<tr>
<td></td>
<td></td>
<td>[\bar{1}10]</td>
</tr>
<tr>
<td>\langle 111 \rangle</td>
<td>Twist</td>
<td>[111]</td>
</tr>
<tr>
<td></td>
<td></td>
<td>[\bar{1}1\bar{1}]</td>
</tr>
</tbody>
</table>

Cartesian space to be accounted for.

\[
\chi = \sqrt{(q_0^2 + q_3^2)(q_1^2 + q_2^2)}
\]

\[
\varphi_1 = \text{atan2} \left( \frac{q_0 q_2 + q_1 q_3}{\chi}, \frac{q_0 q_1 - q_2 q_3}{\chi} \right)
\]

\[
\Phi = \text{atan2} \left( 2\chi, q_0^2 + q_3^2 - q_1^2 - q_2^2 \right)
\]

\[
\varphi_2 = \text{atan2} \left( \frac{q_1 q_3 - q_0 q_2}{\chi}, \frac{q_0 q_1 + q_2 q_3}{\chi} \right)
\]

Inputting the Euler angles into Equation (3.3) created the orientation matrices for the grains. Appendices D and F provide the codes used to generate the orientation matrices.
Testing The Matrices

This work attempted to reproduce the 1D subset graphs as shown in Bulatov et al. as a way to test the different methods. Different methods experienced various levels of success. Figures C.1 to C.3 show the matrices giving the best results.

While the first method works well for MARMOT, the MATLAB® script does not necessarily expect the same GB normal assumed by MARMOT. Thus, the results coming from using this combination of matrices ended up working only for the \( \langle 100 \rangle \) tilt, \( \langle 110 \rangle \) tilt, and \( \langle 100 \rangle \) twist subsets. The \( \langle 110 \rangle \) twist subset had issues with singularities, and the \( \langle 111 \rangle \) subsets did not remotely match the expected outcome.

3.4.2 Calculating Reduced Chi Squared

This work used two methods to calculate the \( \chi^2_{\text{red}} \) statistic. The first method used the P and Q matrices as developed above to test the entirety of the fit. The second method calculated the statistic for each 1D subset, then calculated the full \( \chi^2_{\text{red}} \) value using the statistics from the subsets. Chapter 4 discusses the results from these calculations.

The test for the entire fit used the P and Q matrices to calculate the energy in 1° intervals for each subset, using Bulatov et al.’s GB5DOF.m script. This work used Equation (3.8) to calculate the \( \chi^2_{\text{red}} \) value for each subset and for the entire fit, producing the results in Table 4.1 under the 800 K anneal column under the “\( \chi^2_{\text{red}} \) using P and Q matrices” section,

\[
\chi^2_{\text{red}} = \frac{1}{N - n - 1} \sum_s \frac{(\epsilon_{\text{md}} - \epsilon)^2}{\epsilon_{\text{md}}}. \tag{3.8}
\]

In this equation, \( N \) is the number of observations, \( n \) is the number of parameters, \( \epsilon_{\text{md}} \) are the energies from MD, \( \epsilon \) are the energies from the model, and \( \epsilon \) is the uncertainty in the MD results.

Using the second method, the same angles used in the fitting procedure were used in the RSW equations that create the 1D subsets. The differences between the values resulting
from there and the MD simulation values lead to the $\chi^2_{\text{red}}$ values shown in the 800 K anneal column under the “$\chi^2_{\text{red}}$ comparing the 1D fits” section. This work implemented the same methods to calculate the $\chi^2_{\text{red}}$ values for the data without an anneal.

The statistic calculated using these methods differs from the $\chi^2$ statistic used in the grid-search function to calculate the fitting parameters. The grid-search function used Equation (3.9), and generated values of the same order as Equation (3.8).

$$\chi^2 = \sum (E_{\text{measured}} - E_{\text{calculated}})^2$$  \hspace{1cm} (3.9)
Chapter 4

Results and Discussion

4.1 Validation of P and Q Matrices

Comparison of the energy profiles calculated from the P and Q matrices with the copper energy profiles expected from the parameters defined in Bulatov et al.’s code provides a way to validate the generated matrices. Figure 4.1 shows the results from this comparison for the (100) set, with all six subsets shown in Figures C.1 to C.3. The calculated energies match exactly the predicted values for all but a few points. Each data set does not match the expected energy at 1°, and the tilt data sets also see this mismatch at their second to last data point.

4.2 Fitting Results

Figures 4.2 to 4.4 compare the one-dimensional (1D) results from Harbison and this work. The results show a general decrease in the grain boundary (GB) energies, allowing trends in the different subsets to emerge. These trends allow for an all around better fit, but also introduce some unexpected results. Appendix A shows the parameters calculated from the fitting procedure.

Initial MD recalculations of the (100) symmetric tilt GB energies using the 800 K anneal
Figure 4.1 The $\langle 100 \rangle$ twist (a) and tilt (b) results for the P and Q matrices as compared to Bulatov et al.’s energy profiles. Bulatov et al.’s GB5DOF.m MATLAB® script calculated the expected values by using the default parameters. The GB5DOF.m script calculated the values using the generated matrices. With the exception of the data points at 1° in both (a) and (b) and 89° in (b), the energies calculated from the matrices matches the expected curves exactly.

(Figure 4.2b) showed an unexpected deep cusp around 28°. An analysis of the molecular dynamics (MD) simulation results for this misorientation revealed that, in this case, abnormally high pressures had caused the two crystals to realign. This realignment caused the misorientation angle to change, causing the GB energy to be much lower than expected. Comparison with Harbison’s simulation result revealed that the crystal structure from his simulation did not realign. While Harbison’s did not use annealed data and thus may not represent a global minimum, the data point follows the surrounding data’s trend, justifying the use of his result.

Of the symmetric tilt GB energy sets, the $\langle 110 \rangle$ set has the most improvement. All three sets showed a general decrease in the energy, increasing confidence in the accuracy of the fit for GB energies in uranium dioxide (UO$_2$). However, each of these sets provides more opportunity for research. The $\langle 100 \rangle$ set needs more work done for data points after around 50°. The scatter associated with those points seems to be higher, and the possibility of a slight cusp presents itself around 68°. The $\langle 110 \rangle$ set as mentioned shows the most
improvement, but some low points in the second and third “humps” do not follow the trend, indicating further possibility for cusps. The first part of the function (the first hump) needs additional data to determine the possibility of a cusp between 40° and 50°. The fitted curve to the (111) set now has an unexpected upward trend. The relatively high scatter associated with these data points leads to the possibility of a completely different set of functions to define this subset, meaning additional RSW functions would be required for example.

The twist GB energy sets vary in their success. The (100) set shows little difference between Harbison’s work and this work. An unexpected slight positive concavity at the end of the fitting for this subset indicates the possibility of a cusp. This cusp may occur around 30°. The (110) set has a definite decrease in the overall energies, creating a plateau profile. An additional cusp around 40° might improve the fit. The (111) set has the least improvement. Based on Bulatov et al.’s work,9 this work expected to see a plateau as Harbison’s fitting demonstrated.4 Instead, the fitting produced a curved energy profile, indicating the potential for at least one cusp, possibly around 28°. Preliminary work has changed the number of parameters in an effort to maximize the quality of the fit with a minimal number of parameters. Figure 4.6 compares the current fitting to the tentative new fitting for three of the six 1D subsets. These modified fits in general seem to fit better at the cost of additional parameters, with a smaller $\chi^2_{\text{red}}$ value. Still more parameters may be needed to accommodate additional cusps however. A Levenberg-Marquardt MATLAB® script calculated these tentative fits.21

Figure 4.5 shows the comparison between the values calculated from the P and Q matrices and the expected values from the MD calculations for the (100) subset. Figures C.4 to C.6 shows all six subsets. The (100) tilt subset has an unsolved scaling issue. Overall, the results from the P and Q matrices match the fitted values, with a few anomalies needing to be addressed.
Figure 4.2 The (100) twist (a) and tilt (b) results. In general the re-calculated energies are lower, with significant differences around 40° to 50° in the tilt subset. The unexpected positive concavity in the twist subset around 40° may indicate the presence of a missing cusp. Possible cusps exist around 30° in the twist subset, and around 68° in the tilt subset.

Figure 4.3 The (110) twist (a) and tilt (b) results. Both subsets have significant decreases in energy. The twist subset has a possible cusp at around 40°, and the tilt subset has possible cusps around 40°, 90°, and 140°.

4.3 Reduced Chi-square Results

The $\chi^2_{\text{red}}$ values are much smaller than one for every data set regardless of the method used to calculate the statistic, with the exception of the (100) symmetric tilt subset using the P and Q matrices. This subset has a high $\chi^2_{\text{red}}$ value due to the scaling issue. Because of the
4.3 Reduced Chi-square Results

Figure 4.4 The $\langle 111 \rangle$ twist (a) and tilt (b) results. Most energies are found to be lower, but some are found to be higher. The unexpected positive concavity present in these results could indicate the presence of one or more cusps, with one possible location around $33^\circ$. This work needs additional data to determine possible cusp locations for the tilt subset.

Figure 4.5 A comparison of the expected value of the fitted function with the values calculated using the P and Q matrices for the $\langle 100 \rangle$ 1D tilt subset, with MD values shown for reference. The cause of the scaling issue remains unknown. Low $\chi^2_{\text{red}}$ values, the fitted functions overfit the data. Table 4.1 lists the $\chi^2_{\text{red}}$ values for the 1D subsets using the two different methods for calculation.
Figure 4.6 A comparison of current fitting functions with a possible change to the functions. Dashed lines show the original functions, and the solid lines show the updated functions, with MD results shown for reference. (a) shows a possible change from the Read-Shockley-Wolf (RSW) functions to a simple square root function multiplied by an exponential decay without any theoretical basis. (b) attempts to fit to a cusp around 40°. Further work can be done to find a better fit for this subset. (c) shows the most potential improvement. The potential fit increases the total number of parameters by three to fit to the cusp around 28°. A quick glance at the MD values compared to the fit shows a great improvement from the current fit. To create the graph shown in (c) this work spliced two additional RSW functions into the original function. This created a total of four RSW functions for this subset.
4.3 Reduced Chi-square Results

Table 4.1 A list of the $\chi^2_{\text{red}}$ results using two different methods: using the P and Q matrices for the various orientations to test the fit, and comparing the results of the 1D fits to the 1D data. The values for $\chi^2_{\text{red}}$ are all less than one with the exception of the $\langle 100 \rangle$ symmetric tilt using the P and Q matrices. These values indicate an over-fit to the data.

<table>
<thead>
<tr>
<th>1D Subset</th>
<th>$\chi^2_{\text{red}}$ using P and Q matrices</th>
<th>$\chi^2_{\text{red}}$ comparing the 1D fits</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>No Anneal</td>
<td>800 K Anneal</td>
</tr>
<tr>
<td>$\langle 100 \rangle$ Twist</td>
<td>0.0953</td>
<td>0.1074</td>
</tr>
<tr>
<td>$\langle 110 \rangle$ Twist</td>
<td>0.1010</td>
<td>0.1874</td>
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<tr>
<td>$\langle 111 \rangle$ Twist</td>
<td>0.3041</td>
<td>0.1139</td>
</tr>
<tr>
<td>$\langle 100 \rangle$ Tilt</td>
<td>0.1038</td>
<td>8.7702</td>
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<tr>
<td>$\langle 110 \rangle$ Tilt</td>
<td>4.9799</td>
<td>0.3277</td>
</tr>
<tr>
<td>$\langle 111 \rangle$ Tilt</td>
<td>0.1566</td>
<td>0.7814</td>
</tr>
<tr>
<td>Overall $\chi^2_{\text{red}}$</td>
<td>1.7652</td>
<td>1.4893</td>
</tr>
</tbody>
</table>
Chapter 5

Conclusion

This work has successfully created a more accurate interpolation function for grain boundary (GB) energies in uranium dioxide (UO$_2$), but revealed additional characteristics of the full five-dimensional (5D) GB space that require further research. GB energies were found to be lower when calculated with an 800 K anneal as expected when compared to the data set created without an anneal. Descriptions of those additional characteristics through the use of additional functions will prove beneficial.

Future work should focus on calculating additional data points for fitting. Increasing the number of data points will improve the quality of the fit. Additional data points will also help to identify trends that do not readily appear with the limited data currently available.

Further work could generalize this function to all polycrystalline materials with the fluo-rite crystal structure. Such a generalization would provide further validation of this model, however, such validation would require a sufficient number of GB energies for many different materials with such a structure. The literature does not provide these energies, indicating a need for additional computational resources.

As the model develops, initial conditions set by actual nuclear fuel data will be put into a MARMOT simulation. After simulating grain growth in a nuclear reactor, the fuel data will be compared with the simulation data to determine the accuracy of the model.
Chapter 5  Conclusion
Bibliography


2 Y. Zhang, *Unpublished*, Personal Communication

3 E. Hansen, *Unpublished*, Personal Communication


9 V. V. Bulatov, B. W. Reed and M. Kumar, *Grain boundary energy function for fcc metals*, Acta Mater. 65 (2014), pp. 161–175


## Appendix A

### List of Parameters

**Table A.1** This table gives the parameters for UO$_2$ that generate the energy profiles.

<table>
<thead>
<tr>
<th>Array number</th>
<th>Parameter name</th>
<th>Parameter value</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>Energy Scaling Factor ($e_{RGB}$)</td>
<td>1.6012 J/m$^2$</td>
</tr>
<tr>
<td>2</td>
<td>$\langle 100 \rangle$ Max Distance</td>
<td>0.405</td>
</tr>
<tr>
<td>3</td>
<td>$\langle 110 \rangle$ Max Distance</td>
<td>0.739</td>
</tr>
<tr>
<td>4</td>
<td>$\langle 111 \rangle$ Max Distance</td>
<td>0.352</td>
</tr>
<tr>
<td>5</td>
<td>$\langle 100 \rangle$ Weight</td>
<td>85.3</td>
</tr>
<tr>
<td>6</td>
<td>$\langle 110 \rangle$ Weight</td>
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</tr>
<tr>
<td>7</td>
<td>$\langle 111 \rangle$ Weight</td>
<td>0.08</td>
</tr>
<tr>
<td>8</td>
<td>$\langle 100 \rangle$ Tilt/Twist Mix Power Law (1)</td>
<td>0.03325</td>
</tr>
<tr>
<td>9</td>
<td>$\langle 100 \rangle$ Tilt/Twist Mix Power Law (2)</td>
<td>0.00053125</td>
</tr>
<tr>
<td>10</td>
<td>Maximum $\langle 100 \rangle$ Twist Energy</td>
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</tr>
<tr>
<td>11</td>
<td>$\langle 100 \rangle$ Twist Shape Factor</td>
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<tr>
<td>12</td>
<td>$\langle 100 \rangle$ Asymmetric Tilt Interpolation Power</td>
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<tr>
<td>13</td>
<td>$\langle 100 \rangle$ Symmetric Tilt First Peak Energy</td>
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<table>
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<th>Parameter value</th>
</tr>
</thead>
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<td>14</td>
<td>(100) Symmetric Tilt First Σ5 Energy</td>
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<tr>
<td>15</td>
<td>(100) Symmetric Tilt Second Peak Energy</td>
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<tr>
<td>16</td>
<td>(100) Symmetric Tilt Second Σ5 Energy</td>
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</tr>
<tr>
<td>17</td>
<td>(100) Symmetric Tilt Σ17 Energy</td>
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</tr>
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<td>(100) Symmetric Tilt First Peak Angle</td>
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<td>(100) Symmetric Tilt SecondPeak Angle</td>
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<td>(110) Tilt/Twist Mix Power Law (1)</td>
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<tr>
<td>21</td>
<td>(110) Tilt/Twist Mix Power Law (2)</td>
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<td>(110) Twist Peak Angle</td>
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<td>23</td>
<td>(110) Twist Peak Energy</td>
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<td>(110) Asymmetric Tilt Shape Factor</td>
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<td>(110) Symmetric Tilt Σ3 Energy</td>
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<td>(110) Symmetric Tilt Σ11 Energy</td>
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<td>(110) Symmetric Tilt First Peak Energy</td>
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<td>(110) Symmetric Tilt Second Peak Angle</td>
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<td>34</td>
<td>(110) Symmetric Tilt First Peak Angle</td>
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<td>36</td>
<td>(111) Twist Shape Factor</td>
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<tr>
<td>37</td>
<td>(111) Twist Peak Angle</td>
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<th>Parameter value</th>
</tr>
</thead>
<tbody>
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<td>38</td>
<td>$\langle 111 \rangle$ Twist Peak Energy</td>
<td>0.7971</td>
</tr>
<tr>
<td>39</td>
<td>$\langle 111 \rangle$ Symmetric Tilt Peak Angle</td>
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<td>$\langle 111 \rangle$ Symmetric Tilt Max Energy</td>
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<td>$\langle 111 \rangle$ Symmetric Tilt $\Sigma 3$ Energy</td>
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<td>43</td>
<td>$\langle 111 \rangle$ Asymmetric Tilt Scale Factor</td>
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Appendix A  List of Parameters
Appendix B

Grain Boundary Representations

Visual representations of the GB space helped Bulatov et al. develop their 5D function. However, the size of the five-space in which GBs reside makes representing them difficult. Researchers have developed different methods to represent them, each with their advantages and disadvantages. Three of these methods are the axis-angle representation, the Rodrigues representation, and the fundamental zone representation. These methods, though described separately, can be used together to form a better picture of what the GB space looks like (see for example Figure 3.2b which combines the Rodrigues representation and the fundamental zone representation).

B.1 Axis-Angle Representation

Of the three described, the axis-angle representation most simplistically describes GB space. The axis of rotation of the GB specifies the point in axis-angle space, and the angle of misorientation between the two grains at the GB specifies the magnitude of the vector. Thus, the axis ($\mathbf{a}$) and the angle ($\theta$) mathematically represent an axis-angle vector as:

$$A = \mathbf{a} \theta$$  \hspace{1cm} (B.1)

The axis-angle space can only take into account three degrees of freedom: the two angles
specifying the axis, and the angle rotated through. Thus, axis-angle space cannot fully visualize all of the necessary information contained in the full 5D space. This representation suffers from the difficulties of understanding an infinite space because it maps an axis and an angle onto a Cartesian coordinate system. Without the help of additional methods, this infinite space remains difficult to understand. The best uses of this representation focus on using it as a starting point to move to other, more robust representations, and to represent the misorientation between two grains.\cite{17}

### B.2 Rodrigues Representation

The Rodrigues representation (sometimes called the “Rodrigues-Frank” representation) uses Rodrigues vectors to represent rotations in Rodrigues space. This representation takes ideas from the axis-angle space, but makes a few changes allowing crystal symmetries to be taken into account. The orientation of the GB normal still specifies the point in space, but the tangent of half the angle represents the magnitude of the vector. Thus, a Rodrigues vector can be represented as: \cite{17,22-25}

\[
\mathbf{R} = a \tan \left( \frac{\theta}{2} \right) \tag{B.2}
\]

Some researchers favor this representation over others because of the lack of curvature such a mapping entails.\cite{17,22} However, it still only specifies three of the five degrees of freedom. Bulatov et al. attached a unit vector at the points along the axis to represent the other two DoFs in Figure 3.2b. A parallel vector represents a twist boundary, and a perpendicular vector represents a tilt boundary. Anything else represents a mix of twist and tilt (or a mixed boundary). One limitation of Rodrigues space lies in that it also maps to an infinite space.\cite{22,26}
B.3 Fundamental Zone Representation

The fundamental zone graphically represents the full 5D GB best. This representation takes advantage of the symmetries inherent in crystals\textsuperscript{13} to simplify an infinite space into a compact, finite area called the fundamental zone.\textsuperscript{6,8,9,23,27} Every point within the space represents a unique orientation, and every point outside the space can be represented as a point inside the space through symmetry operations.\textsuperscript{22–24} Bulatov \textit{et al.} used this idea in connection with Rodrigues space to create Figure 3.2b. In Rodrigues space, the crystal symmetries of the material determine the shape of the fundamental zone.\textsuperscript{6,23} For fcc crystals, the fundamental zone takes the form of a truncated tetrahedron.\textsuperscript{9} The edges of the fundamental zone in Rodrigues space represent the high-symmetry rotation axes, and points on one face can represent another point on a different face of the fundamental zone.
Appendix C

Graphs

Figure C.1  The (100) twist (a) and tilt (b) results for the P and Q matrices as compared to Bulatov et al.’s energy profiles. Bulatov et al.’s GB5DOF.m MATLAB® script calculated the expected values by using the default parameters. The GB5DOF.m script calculated the values using the generated matrices. With the exception of the data points at 1° in both (a) and (b) and 89° in (b), the energies calculated from the matrices matches the expected curves exactly.
Figure C.2 The $\langle 110 \rangle$ twist (a) and tilt (b) results for the P and Q matrices as compared to Bulatov et al.’s energy profiles. Bulatov et al.’s GB5DOF.m MATLAB® script calculated the expected values by using the default parameters. The GB5DOF.m script calculated the values using the generated matrices. With the exception of the data points at $1^\circ$ in both (a) and (b) and $179^\circ$ in (b), the energies calculated from the matrices matches the expected curves exactly.

Figure C.3 The $\langle 111 \rangle$ twist (a) and tilt (b) results for the P and Q matrices as compared to Bulatov et al.’s energy profiles. Bulatov et al.’s GB5DOF.m MATLAB® script calculated the expected values by using the default parameters. The GB5DOF.m script calculated the values using the generated matrices. With the exception of the data points at $1^\circ$ in both (a) and (b) and $60^\circ$ in (b), the energies calculated from the matrices matches the expected curves exactly.
Figure C.4  A comparison of the expected value of the fitted function with the values calculated using the P and Q matrices for the \(\langle 100\rangle\) 1D subsets, with the MD values shown for reference. (a) PQ results follow exactly the fitted curve. (b) has a scaling issue yet to be fixed. The cause of the scaling issue remains unknown.

Figure C.5  A comparison of the expected value of the fitted function with the values calculated using the P and Q matrices for the \(\langle 110\rangle\) 1D subsets, with the MD values shown for reference. (a) follows the fitted result until the cusp, at which point some anomalies appear. The results from the PQ matrices dip well below the expected value at the cusp, and never make it back to the original fitted line. (b) has a similar issue on a lesser scale. Only two of the calculated points do not follow the fitted curve. At the endpoint the expected value is zero, but the PQ matrices calculated a value slightly higher. Also, an unexpected cusp from the PQ matrices appears in the middle of the second hump. All other data points follow the fitted curve exactly.
Figure C.6 A comparison of the expected value of the fitted function with the values calculated using the P and Q matrices for the (111) 1D subsets, with the MD values shown for reference. (a) closely follows the expected fitted values, but has a slight error throughout. (b) follows the expected values exactly in the center of the fitting, but misses slightly for lower angle boundaries, and misses completely at the end.
Appendix D

Orientation Matrix Generator

This code generates the orientation matrices (known as the P and Q matrices in Bulatov et al.’s code). Provision for calculating the matrices one of two ways appears in-code through the use of command-line options.

```python
from __future__ import division,print_function # To avoid numerical problems with division, and for ease of printing
from sys import argv # for CLI arguments
from math import cos, sin, pi, atan2, sqrt # Trig functions
from os.path import exists # For checking existence of a file
from numpy import array, linalg
from myModules import * # imports my functions from the file myModules.py

# Helper functions
def displayHelp():
    print('This script will calculate the orientation matrices for any given misorientation for any of the high-symmetry axes.
Arguments:

_axis: The axis of orientation (type: int)

')
```

47
Appendix D  Orientation Matrix Generator

_misorientation: The angle of misorientation (type: float)

--------OR--------
(with option -e or --euler)

_z1: The first rotation angle (Z) (type: float)
_x: The second rotation angle (X') (type: float)
_z2: The third rotation angle (Z") (type: float)

If the option -e or --euler-angles is entered, the calculation skips
to simply
output the orientation matrices. Otherwise, the Euler angles are
calculated from
the axis, orientation, and grain boundary normal, and then the
orientation matrix is
created through the use of the Rodrigues Rotation Formula, which is:

$R = I + \sin(\theta) \times K + (1 - \cos(\theta)) \times K^2$

where I is the identity matrix, theta is the misorientation angle, and
K is
the skew-symmetric matrix formed by the axis of rotation:

$K = \begin{bmatrix}
0 & -kz & ky \\
-ky & 0 & -kx \\
kz & kx & 0
\end{bmatrix}$

where the vector k is the unit vector defining the axis of rotation,
or using
a set of predefined rotations for each axis (default is the predefined
rotations).
The Euler angles are calculated in this case simply for the file to be
written
to. If the user does not specify to save, then the angles are not
used for
anything.

Options:
-e --euler <_z1> <_x> <_z2>
  Returns the Bunge orientation matrix based on the euler angles provided.

-f --file <filename>
  Reads the file filename and uses Euler angles from them to calculate the
  orientation matrix.

--rrf
  Calculates the matrices using the Rodrigues Rotation Formula.

-a --angles
  Displays the Euler angles. Can be used in conjunction with -q or --quiet to
display only the Euler angles.

-s --save
  Saves the resultant orientation matrix to a database (orientation_matrix_database.m)
  with the accompanying Euler angles.

-q --quiet
  Suppresses output of the orientation matrices to the terminal.
Appendix D  Orientation Matrix Generator

Output:
For an Euler angle set, the output is simply its orientation matrix.
For the misorientations, the first matrix is the 'P' orientation matrix, and
the second matrix is the 'Q' orientation matrix (see Bulatov et al.,
Acta Mater
65 (2014) 161-175).

def displayAngles(z1, x, z2): # Displays an Euler angle set (Bunge convention)
    print("Euler angles:")
    # This is the "new" way to format strings. The 16 indicates the
    # be done before the next character. The '<' character below says
    # to pad (the right side).
    print("{:16}{:16}{:16}".format('Z', 'X', 'Z'))
    print("----------------------------------------")
    print("{: <16}{: <16}{: <16}").format(rad2deg(z1), rad2deg(x), rad2deg(z2)))
    return

def check4RRF(args): # Check the args for the rrf command
    if "--rrf" in args:
        index = args.index("--rrf")
        del args[index]
        return True, args
    else:
        return False, args
def check4Euler(args): # Check the args for the -a or --angles command
    if "-a" in args or "--angles" in args:
        try:
            index = args.index("-a")
        except:
            index = args.index("--angles")
        del args[index]
        return True, args
    else:
        return False, args

# Write the matrix and angles to a file
def writeMat(m, _z1, _x, _z2, grain, axis):
    # This is to avoid issues with duplicates
    if _z1 == 0:
        _z1 = abs(_z1)
    if _x == 0:
        _x = abs(_x)
    if _z2 == 0:
        _z2 = abs(_z2)

    lastVal = 1
    # This is the default filename to be used.
    # TODO: make provisions to provide the database file via command line
    tex_filename = "orientation_matrix_database.m"
    var_name = "%s%d"%(grain, axis) # Will generally look like P100 or Q100
    if not exists(tex_filename):
        tex_file = open(tex_filename, "a")
Appendix D  Orientation Matrix Generator

tex_file.write("% Database for orientation matrices for specified Euler Angles
")

tex_file.write("%-------------------------------------------------------------------\n")

tex_file.write("% Orientation Matrix
")

tex_file.write("%s(: ,: ,%d) =[%2.6f %2.6f %2.6f %%%2.4f %2.4f %2.4f \n"%(var_name , lastVal , m[0][0] , m[0][1] , m[0][2] ,
_z1 , _x , _z2 ))

tex_file.write("%2.6f %2.6f %2.6f \n"%(m[1][0] , m[1][1] , m[1][2] ,
)

tex_file.write("%2.6f %2.6f %2.6f ]; \n"%(m[2][0] , m[2][1] , m[2][2] )

)

tex_file.write("%-------------------------------------------------------------------\n")

tex_file.close()

else:

f = open(tex_filename,"r")

while True:

  data = f.readline().split()

  if not data:

    break

  elif len(data) != 6:

    continue

  else:

    assert data[0][0] in {'P', 'Q'}, "Unknown orientation matrix type (should be \'P\' or \'Q\')."

    if not "%d"%(axis) in data[0][1:4]:

      lastVal = 0

    elif "%d"%(axis) in data[0][1:4]:


try:

    try:
        if data[0][0] == 'P':  # Handles anything 3 digits long
            lastVal = int(data[0][9:12]) - 1
        else:
            lastVal = int(data[0][9:12])
    except:
        if data[0][0] == 'P':  # Handles anything 2 digits long
            lastVal = int(data[0][9:11]) - 1
        else:  # data[0][0] == 'Q'
            lastVal = int(data[0][9:11])

else:

    print("Error: Unknown last index.")
    exit()

# Checks to see if the Euler angles have already been used before
# If so, the calculated matrix is not saved (assumed to already
# be in the database)
if data[0][0] == grain and data[3] == ('%' + "%2.4f"%_z1)
    and data[4] == "%2.4f"%_x and data[5] == "%2.4f"%_z2:
    unique = False
    break
else:
unique = True

if unique:
    tex_file = open(tex_filename, "a")
    tex_file.write("%s(:,:,%d)=[%2.6f %2.6f %2.6f
        %%%2.4f %2.4f %2.4f
"%(var_name, lastVal + 1, m[0][0], m
[0][1], m[0][2], _z1, _x, _z2))
    tex_file.write("%2.6f %2.6f %2.6f\n"%(m[1][0], m[1][1], m
[1][2]))
    tex_file.write("%2.6f %2.6f %2.6f
")
    tex_file.write("%-------------------------------------------------------------------\n"
    tex_file.close()

return

if "--help" in argv:  # Help info
displayHelp()
exit()

orientation_matrix = []

save, argv = check4Save(argv)  # Save the file? Delete the save argument
quiet, argv = check4Quiet(argv)  # Checks for suppressing output. Delete
    the quiet argument.
useRRF, argv = check4RRF(argv)  # Checks for using the RRF method. Delete
    the rrf argument.
dispEuler, argv = check4Euler(argv)  # Checks for displaying the Euler
    angles. Delete the angle argument

# If the arguments come from a file...
if "-f" in argv or "--file" in argv:  # input arguments come from file
    try:
try:
    index = argv.index("-f")
except:
    index = argv.index("--file")

except:
    print("ERROR: Unable to find filename.")
    exit()

filename = argv[index + 1]

try:
    f1 = open(filename, 'r')
except:
    print("ERROR: Unable to read file.", filename)

while True:  # Read the file line by line.
    line = f1.readline()
    # break if we don't read anything. If there are blank lines in the
    # file, this will evaluate to TRUE!
    if not line:
        break;
    data = line.split()

    if len(data) != 4:  # If there are less than 4 parts to the data,
        # move along (format of file MUST be _z1 _x _z2 1.00)
        continue
    else:
        # Convert the data to stuff we can use
        _z1 = float(data[0])
        _x = float(data[1])
        _z2 = float(data[2])

        _z1 = deg2rad(_z1)
        _x = deg2rad(_x)
_z2 = deg2rad(_z2)
orientation_matrix = calcRotMat(_z1, _x, _z2)

if not quiet:
displayMat(orientation_matrix)

if save:
writeMat(orientation_matrix, _z1, _x, _z2, 'P', _axis)

# Input is a set of euler angles
elif '-e' in argv or '--euler-angles' in argv:
    try:
        try:
            index = argv.index('-e')
        except:
            index = argv.index('--euler-angles')
    except:
        print("ERROR: Unable to read Euler angles.")
        exit()

    _z1 = float(argv[index + 1])
    _x = float(argv[index + 2])
    _z2 = float(argv[index + 3])

    _z1 = deg2rad(_z1)
    _x = deg2rad(_x)
    _z2 = deg2rad(_z2)

    orientation_matrix = calcRotMat(_z1, _x, _z2)

    if not quiet:
        displayMat(orientation_matrix)

    if save:
        writeMat(orientation_matrix, _z1, _x, _z2, 'P', _axis)

else:
if len(argv) < 3:
    print("ERROR: Not enough command line arguments.")
    displayHelp()
    exit()

try:
    _axis = int(argv[1])
    _misorientation = float(argv[2])
except:
    print('ERROR: Command line argument(s) is (are) not of correct type. Please enter an int for argument 1, a float for argument 2, and an int for argument 3.
    ",
    exit()

if not len(str(_axis)) == 3:  # axis length greater than 3
    print("ERROR: Argument 1 must by a 3 digit number like '100'.")
    exit()

_misorientation = deg2rad(_misorientation)  # Change input to radians
axis = [None]*3
_z1 = [None]*2
_x = [None]*2
_z2 = [None]*2
q = [None]*2
for i in range(0, len(str(_axis))):
    axis[i] = int(str(_axis)[i])
# The Actual Calculations

# First convert to a quaternion
# These functions are from a myModules.py.
q[0] = axis2quat(axis, _misorientation / 2)
q[1] = axis2quat(axis, -_misorientation / 2)

# Convert the quaternion to Euler Angles
for i in range(0, len(_z1)):
    _z1[i], _x[i], _z2[i] = quat2euler(q[i])

# Using the Rodrigues Rotation Formula, defined as R = I + sin(theta)
# K + (1 - cos(theta))*K^2
# with K = [0 -k_z, k_y; k_z, 0, -k_x; -k_y, k_x, 0], and the
# components of
# k coming from the vector being rotated about. Theta is specified by
# the misorientation.
if useRRF:
    orientation_matrix1, orientation_matrix2 = calcRotMatRRF(axis,
                                                              _misorientation)

    # Normalize the matrices using their determinants
    orientation_matrix1 = orientation_matrix1 / linalg.det(orientation_matrix1)
orientation_matrix2 = orientation_matrix2 / linalg.det(orientation_matrix2)

if not quiet:
    displayMat(orientation_matrix1)
    displayMat(orientation_matrix2)

for i in range(0, len(_z1)):
    if dispEuler:
        displayAngles(_z1[i], _x[i], _z2[i])

if save:
    assert i < 2, "ERROR: Too many Euler angles."
    if i == 0:
        writeMat(orientation_matrix1, _z1[i], _x[i], _z2[i], 'P', _axis)
    else:
        writeMat(orientation_matrix2, _z1[i], _x[i], _z2[i], 'Q', _axis)

else:
    for i in range(0, len(_z1)):
        orientation_matrix = calcRotMat(_z1[i], _x[i], _z2[i])

        # Normalize the matrix using the determinant
        orientation_matrix = orientation_matrix / linalg.det(orientation_matrix)

        if not quiet:  # Display the results
            displayMat(orientation_matrix)
if dispEuler:  # Display the Euler Angles
displayAngles(_z1[i], _x[i], _z2[i])

if save:  # We only calculate 2 angles at a time. If there are more, that’s a problem.
    assert i < 2, "ERROR: Too many Euler angles."

    if i == 0:
        writeMat(orientation_matrix, _z1[i], _x[i], _z2[i], 'P', _axis)
    else:
        writeMat(orientation_matrix, _z1[i], _x[i], _z2[i], 'Q', _axis)
This code generates the rotation matrices used to rotate the axes to the [100] direction as required by Bulatov et al.’s script.

```python
from __future__ import division, print_function # Automatically divides as floats, and considers print() a function
from sys import argv # for CLI arguments
from numpy import array, linalg # for matrix operations
from os.path import exists # For checking existence of a file
from myModules import * # For using my user-defined functions

# Helper functions

def displayHelp():
    print('''
This script will generate the rotation matrix for the given misorientation axis
Input:

_rotation_axis This specifies the axis around which the grains are rotated. (type: int (100) or string ('100'))
'''
)
```

Input:

_rotation_axis This specifies the axis around which the grains are rotated. (type: int (100) or string ('100'))
_gbnormal

This specifies the boundary plane normal. (type:

int (100) or string ('100'))

Options:

-s --save Saves the resultant rotation matrix to a
database

(rotation_matrix_database.m) with the
accompanying
rotation axis and misorientation type.

-q --quiet Suppresses output of the rotation matrix

--help Display this help info.

Output:
The output displayed will be the resultant rotation matrix for the
given
misorientation.

# This function is an adaptation from MOOSE RotationMatrix class.
def rotVecToZ(vec): # Creates the rotation matrix to rotate vec to the z
direction

    # REALLY make sure vec is normalized
    vec = vec / linalg.norm(vec)

    # Initialize our vectors
    v1 = array([[0.,0.,0.]])
    v0 = array([[0.,0.,0.]])

    # Temp vector that gives a prototype of v1 by looking at the smallest
    component of vec
    w = array([[abs(vec[0][0]), abs(vec[0][1]), abs(vec[0][2])]])
if \((w[0][2] \geq w[0][1] \text{ and } w[0][1] \geq w[0][0]) \text{ or } (w[0][1] \geq w[0][2] \text{ and } w[0][2] \geq w[0][0])\):
    v1[0][0] = 1.0
elif \((w[0][2] \geq w[0][0] \text{ and } w[0][0] \geq w[0][1]) \text{ or } (w[0][0] \geq w[0][2] \text{ and } w[0][2] \geq w[0][1])\):
    v1[0][1] = 1.0
else:
    v1[0][2] = 1.0

# Gram-Schmidt method to find \(v_1\)
\(v_1 = v_1 - ((v_1 \cdot vec) \cdot vec)\)
\(v_1 = v_1 / \text{linalg.norm}(v_1)\)

# \(v_0 = v_1 \times vec\)
\(v_0[0][0] = v_1[0][1] \cdot vec[0][2] - v_1[0][2] \cdot vec[0][1]\)
\(v_0[0][1] = v_1[0][2] \cdot vec[0][0] - v_1[0][0] \cdot vec[0][2]\)
\(v_0[0][2] = v_1[0][0] \cdot vec[0][1] - v_1[0][1] \cdot vec[0][0]\)

# Rotation matrix is just:
\(\text{rot} = \text{array}([[v_0[0][0], v_0[0][1], v_0[0][2]],
                   [v_1[0][0], v_1[0][1], v_1[0][2]],
                   [vec[0][0], vec[0][1], vec[0][2]]])\)

return rot

def rotVec1ToVec2(vec1, vec2):
    rot1_to_z = rotVecToZ(vec1)
    rot2_to_z = rotVecToZ(vec2)
    return (rot2_to_z.T).dot(rot1_to_z)

def writeMat(m, _axis, gbnormal): # Write the matrix and angles to a file
tex_filename = "rotation_matrix_database.m"
normName = _axis + '_x' + gbnormal
Appendix E Rotation Matrix Generator

```python
var_name = "rot%snorm"%(normName)

if not exists(tex_filename):
    tex_file = open(tex_filename, "a")
    tex_file.write("% Database for rotation matrices for specified
    misorientation axes\n")
    tex_file.write("%----------------------------------------------------------------\n")
    tex_file.write("% Rotation Matrix\n")
    tex_file.write("%s=[%2.4f %2.4f %2.4f\n"%(var_name, m[0][0], m
    [0][1], m[0][2])
    tex_file.write("%2.4f %2.4f %2.4f\n"%(m[1][0], m[1][1], m[1][2])
    tex_file.write("%2.4f %2.4f %2.4f ];\n"%(m[2][0], m[2][1], m
    [2][2])
    tex_file.write("%----------------------------------------------------------------\n")
    tex_file.close()

else:
    # Check for already written
    numlines = countFileLines(tex_filename)
    if numlines <= 4:
        tex_file = open(tex_filename, "a")
        tex_file.write("%s=[%2.4f %2.4f %2.4f\n"%(var_name, m[0][0], m
        [0][1], m[0][2])
        tex_file.write("%2.4f %2.4f %2.4f\n"%(m[1][0], m[1][1], m
        [1][2])
        tex_file.write("%2.4f %2.4f %2.4f ];\n"%(m[2][0], m[2][1], m
        [2][2])
        tex_file.write("%----------------------------------------------------------------\n")
```

n")
tex_file.close()

else:
    f = open(tex_filename,"r")

while True:
    data = f.readline().split()
    if not data:
        break
    elif len(data[0]) > 14:
        if not data[0][14] == '=':
            continue
        else:
            if data[0][0:10] == var_name:
                unique = False
            else:
                unique = True
    if unique:
        tex_file = open(tex_filename, "a")
        tex_file.write("%s=[%2.4f %2.4f %2.4f\n"%(var_name, m[0][0], m[0][1], m[0][2]))
        tex_file.write("%2.4f %2.4f %2.4f\n"%(m[1][0], m[1][1], m[1][2]))
        tex_file.write("%2.4f %2.4f %2.4f ;\n"%(m[2][0], m[2][1], m[2][2]))
        tex_file.write("%-------------------------------------------------------\n")
        tex_file.close()

# Check what we were given...
if "--help" in argv: # Help info
displayHelp()
exit()

save, argv = check4Save(argv)  # Checks for suppressing output

if len(argv) != 3:  # if not both values given
    print("ERROR: Incorrect number of command line arguments. Line 150")
    displayHelp()
    exit()
else:  # len(argv) == 3
    script, _rotation_axis, _gbnormal = argv

    if not type(_gbnormal) in {int, str}:
        print("ERROR: Grain boundary normal type is incorrect. Please enter an int or a string. Line 157")
        print("You entered %s with type %s"%(str(_gbnormal), type(_gbnormal)))
        exit()
    else:
        if type(_gbnormal) == int:
            _gbnormal = '0' + '0' + '0' + str(_gbnormal)
            _gbnormal = _gbnormal[-3:]  # gets the last three characters
            assert _gbnormal != '000', "ERROR: invalid boundary normal. Line 173"

        assert(len(_rotation_axis) == 3), "ERROR: Something went wrong converting _gbnormal into a string. Line 166"

        if not type(_rotation_axis) in {int, str}:
            print("ERROR: Grain boundary rotation axis type is incorrect. Please enter an int or a string. Line 169")
            print("You entered %s with type %s"%(str(_rotation_axis), type(_rotation_axis)))
exit()

else: # Convert anything besides a string into a string
    if type(_rotation_axis) == int:
        _rotation_axis = '0' + '0' + '0' + str(_rotation_axis)
    _rotation_axis = _rotation_axis[-3:] # Get the last three characters
    assert _rotation_axis != '000', "ERROR: invalid rotation axis."

assert(len(_rotation_axis) == 3), "ERROR: Something went wrong converting _rotation_axis into a string."

# Now that the input is taken care of, do the work
axis = array([[1, 0, 0]])  # This is the axis that we rotate the grain boundary normal to

# This part converts _gbnormal to an array for use in the rotation functions

        gbnormal = array([None]*3)
    j = 0
    indices = []
    for i in range(0, len(gbnormal[0])):
        try:
            gbnormal[0][i] = int(_gbnormal[j])
            j = j+1
        except:
            print(_gbnormal[i:i+2])
            gbnormal[0][i] = int(_gbnormal[j:j + 2])
            j = j + 2
    indices.append(i)
# So much work...

gbnorm = ''
for i in range(0, len(gbnormal[0])):
    if gbnormal[0][i] < 0:
        gbnorm += str(abs(gbnormal[0][i])) + 'bar'
    else:
        gbnorm += str(gbnormal[0][i])

gbnormal = gbnormal / linalg.norm(gbnormal) # Normalize the gbnormal vector.

axis = axis / linalg.norm(axis) # Just to be sure...

rotation_matrix = rotVec1ToVec2(gbnormal, axis)

if not quiet:
    displayMat(rotation_matrix)

if save:
    writeMat(rotation_matrix, _rotation_axis, gbnorm)
This bash script reads a CSV file containing misorientation angles data, and uses those angles to generate the P and Q matrices. This script calls the script orientation_matrix.py.

```bash
#!/bin/bash

# This script will generate the orientation matrices through python by looping
# through the CSV values given in the input files.
# Argument(s):
#   $1 Should be a filename that specifies the angles and relative
#       energies for the 100, 110, and 111 symmetric tilt and twist
# Command-line argument counter that checks for the correct number of arguments.
# Does not check for correct syntax.
if [ "#$" -ne 1 ]; then
  echo "Illegal number of parameters"
  exit 1
fi
```

Appendix F

**genOrientationMatrix.sh Bash Script**

This bash script reads a CSV file containing misorientation angles data, and uses those angles to generate the P and Q matrices. This script calls the script `orientation_matrix.py`.

```bash
#!/bin/bash

# This script will generate the orientation matrices through python by looping
# through the CSV values given in the input files.
# Argument(s):
#   $1 Should be a filename that specifies the angles and relative
#       energies for the 100, 110, and 111 symmetric tilt and twist

# Command-line argument counter that checks for the correct number of arguments.
# Does not check for correct syntax.
if [ "#$" -ne 1 ]; then
  echo "Illegal number of parameters"
  exit 1
fi
```
# This takes the first argument from the command line - this is assumed to be a filename of the format 100Tilt.

FN=$1

echo "Determining the axis..."

# Pulls out the axis from the input file name. This uses regex syntax to find a series of numbers that match either 100, 110, or 111. This also has an issue where it will find a match for 101, but as long as the files are named correctly it shouldn’t be an issue.

 AXIS='echo $FN | grep -o "[101][01]"'

echo "Reading the file..."

IFS="," # separation character is the comma

# Exit with error code 99 if unable to read the file

[ ! -f $FN ] && { echo "$FN file not found"; exit 99; }

# This makes the assumption that the file orientation_matrix.py has executable # rights.

echo "Running the command: "~/projects/scripts/orientation_matrix.py $AXIS
<angle> -s -q"

while read -r angle en; do # read the file with comma separated values

~/projects/scripts/orientation_matrix.py $AXIS $angle -s -q
done < "$FN" # the "$FN" is required if it’s going to run properly!

IFS=$OLDIFS # go back to the old separation character based on the system value.